

Interaction between $\text{La}_{0.7}\text{Sr}_{0.3}\text{MnO}_3$ and SrZrO_3 in CMR composites

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Annealing of pressed mechanical mixtures of finely grained $\text{La}_{0.7}\text{Sr}_{0.3}\text{MnO}_3$ and coarse-grained SrZrO_3 at $T = 1000\text{--}1300\text{ }^\circ\text{C}$ was accompanied by a systematic decrease in SrZrO_3 lattice parameters with increasing $\text{La}_{0.7}\text{Sr}_{0.3}\text{MnO}_3 : \text{SrZrO}_3$ ratio and a corresponding decrease in sample magnetization at room temperature. EDX nanoprobe analysis of the composites confirmed substantial out-diffusion of La and Mn from $\text{La}_{0.7}\text{Sr}_{0.3}\text{MnO}_3$ into the SrZrO_3 particles. This process, detectable at $1100\text{ }^\circ\text{C}$, leads to an almost uniform distribution of La and Mn (6–8%) in the SrZrO_3 grains and the corresponding changes in manganite particle composition at $1300\text{ }^\circ\text{C}$. Zr diffusion backflow is almost absent.

Introduction

The practical application of manganite-based ceramic materials that demonstrate the colossal magnetoresistance (CMR) effect is seriously complicated by their relatively low magnetoresistance (MR) in weak magnetic fields at room temperature. Recent investigations have demonstrated the possibility of substantially enhancing this parameter in manganite–insulator composites, using various insulating components.^{1–4} The cause of this effect seems to be related to the increase of MR in the vicinity of the percolation boundary in these systems. At the same time, the published values for the ratios of the components that result in the CMR enhancement are often rather far from the formal percolation boundaries. This discrepancy can result from chemical interaction between the components during the thermal processing of the composites.

In some of these composites, an intensive reaction between the components is accompanied by the appearance of crystalline reaction products,⁵ but more often XRD does not detect any new phases. This fact is usually claimed as a proof of the absence of any interaction between the components. At the same time such effects as an abrupt increase in resistivity at rather small insulator content⁶ or a shift of components ratio to higher manganite content at higher thermal processing temperatures, corresponding to an MR enhancement,³ clearly show some kind of chemical reaction in the composites during the thermal processing. However, the details of such interactions have not been studied until now.

One of the largest values of CMR enhancement compared to single phase manganite ceramics (3.5% vs. 0.2–0.3% at $H = 100\text{ Oe}$ and $T = 298\text{ K}$)³ was observed for the composites $\text{La}_{0.7}\text{Sr}_{0.3}\text{MnO}_3(\text{LSM})\text{--SrZrO}_3$. The high temperature interaction of these phases has not been studied, but the reaction of LSM with yttria-doped zirconia (YSZ) during the preparation and processing of solid oxide fuel cells (SOFCs) has been investigated in detail by several groups.^{7–13} Depending on the processing conditions and the manganite composition, this reaction results in the appearance of $\text{La}_2\text{Zr}_2\text{O}_7$ and/or SrZrO_3 , which remain unchanged during further thermal processing. The basic mechanism of this reaction consists of La and Sr out-diffusion into the YSZ, leading to the formation of a $\text{SrZrO}_3\text{--La}_2\text{Zr}_2\text{O}_7$ layer at the interface between the LSM and the YSZ.^{7,11} Along with the crystalline reaction product(s), the formation of amorphous product areas was observed.⁸ At

the same time these studies gave no evidence of any interaction between LSM and SrZrO_3 , which could be the reason for the effects observed in the LSM– SrZrO_3 composites.

Experimental

The details of the composite preparation are described elsewhere.³ Preliminary synthesized single phase submicron LSM and pre-annealed micron-sized SrZrO_3 powders were mixed in various ratios (20–80% LSM) in a ball mill in ethanol, they were then pelletized, annealed at various temperatures (1000–1300 °C) for 1 h in air and cooled to room temperature at 5 K min^{-1} . The composite samples obtained were crushed in a mortar and studied by XRD (Cu $K\alpha$, 1° min^{-1} , Si internal standard), VSM magnetometer (room temperature, $H \leq 1.5\text{ T}$) and TEM, equipped with a nanoprobe EDX. Additional calibration of EDX in the nanoprobe mode was performed using crushed LSM and SrZrO_3 ceramics.

Results and discussion

Several anomalies³ of the compositional dependence of the magnetoresistance curves of the LSM– SrZrO_3 composites showed a possible chemical interaction between the LSM and the SrZrO_3 during thermal processing. Initial XRD analysis of the composite samples did not detect the appearance of new phases resulting from a solid state reaction between the components, which correlates quite well with previous results of analysis of LSM–YSZ interaction products.^{8–10} It should also be noted that weak field CMR effects are sensitive to even small compositional fluctuations so that even the primary stages of topochemical reaction, which cannot be detected by usual XRD, can result in significant changes in the magnetic properties of the composite.

More detailed investigation of the SrZrO_3 lattice parameters for the samples having various LSM content and being processed at different temperatures, confirmed the existence of a chemical reaction between the composite components. Increasing the LSM– SrZrO_3 ratio resulted in a monotonous and systematic decrease in both the *a* and *b* parameters of the tetragonal unit cell, clearly visible even at $T = 1100\text{ }^\circ\text{C}$, even though the CMR data indicate no trace of this interaction yet (Fig. 1). In spite of a short reaction time (1 h), a further increase

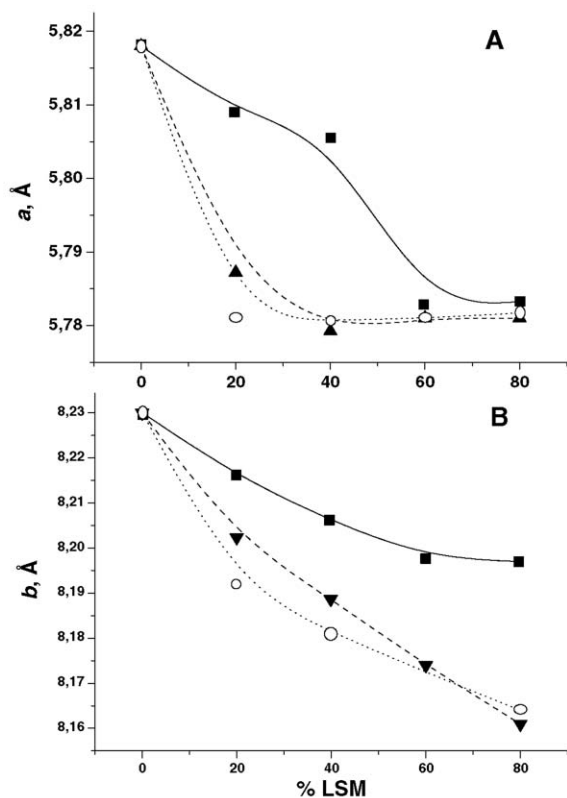


Fig. 1 *a* (A) and *b* (B) lattice parameters of SrZrO₃ unit cell in LSM–SrZrO₃ composites annealed at various temperatures: ■, 1100 °C; ▲, 1200 °C; ○, 1300 °C.

in annealing temperature leads to an abrupt decrease in the *a* parameter even with a minimal LSM content (20%). Due to the negligible self-nonstoichiometry of the strontium zirconate the most probable reason for this might be diffusion of lanthanum and/or manganese into the SrZrO₃ lattice. As mentioned before, the literature^{7–12} did not indicate the presence of any La in the SrZrO₃ that appeared as a reaction product between the LSM and YSZ, while other authors¹³ detected significant amounts of La in Sr-rich regions of the product layer. The solubility limit of La in SrZrO₃, as determined by a solid-state reaction method,¹⁴ is about 6 mol%; a further increase in La content was accompanied by the formation of La₂Zr₂O₇. Substitution of Sr for La resulted in lattice contraction, but the lattice parameter reduction was ~10 times lower than observed in the present work for LSM–SrZrO₃ composites. Hence, the drop in the *a* parameter at 1200–1300 °C cannot be attributed only to La diffusion into the SrZrO₃. Earlier papers mentioned the possibility of Mn diffusion into SrZrO₃,¹⁵ but later investigations did not detect a similar phenomenon. At the same time other authors⁷ reported the intensive diffusion of both La and Mn into YSZ during LSM–YSZ interaction.

If the evolution of SrZrO₃ lattice parameters is related to the La and Mn diffusion, such changes should be accompanied by simultaneous processes in another component of the LSM–SrZrO₃ composite. A precise analysis of the lattice parameters of the lanthanum strontium manganite is complicated by substantial peak broadening, which is especially pronounced at lower annealing temperatures. However, analysis of the magnetic response of the composites obtained, as controlled by the magnetization of the ferromagnetic LSM component, confirmed a substantial evolution of sample magnetization during annealing (Fig. 2). Single-phase LSM ceramics, taken for comparison, demonstrate a monotonous increase in magnetization, in a field of 1.5 T, with increased annealing temperature due to the further crystallographic ordering and grain growth in ferromagnetic La–Sr manganite. At 20% SrZrO₃,

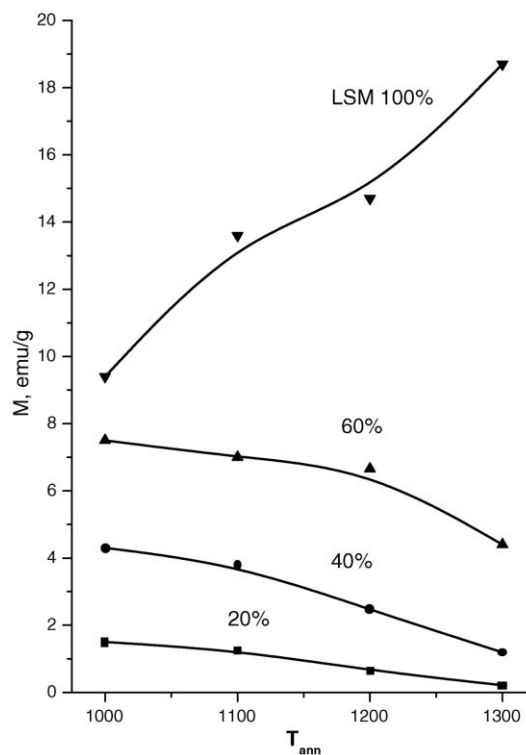


Fig. 2 Magnetization ($H = 1.5$ T) of composite samples, annealed for 1 h at various temperatures.

annealing of the composites at $T > 1100$ °C leads to the reverse effect; after the initial crystallization-related increase, the magnetization manifested a monotonous fall with increasing temperature, which cannot be explained by intrinsic processes within the LSM phase. This process is especially evident at an SrZrO₃ content of >40%, when any kind of annealing results in a progressive fall of magnetization. If the change in SrZrO₃ lattice parameters is caused by La and/or Mn out-diffusion from the LSM, the magnetization fall observed at elevated processing temperatures can be attributed to the shift of Curie temperature of the LSM in a progressively increasing reaction zone. In which case an ambient temperature magnetization measurement should demonstrate a decreasing amount of ferromagnetic phase.

The most direct evidence of the solid-state chemical interaction of the two phases can be given by local composition analysis. Previous studies of similar interactions have been limited by the size of the EDX microprobe excitation area (1–3 microns), which exceeds the size of the interacting objects in many cases. Application of a nanoprobe with an excitation zone of 15–20 nm provides the possibility of measuring the element distribution within the submicron particles. The analysis of the LSM and SrZrO₃ particles contact area after annealing at 1100 °C (Fig. 3a) allows us to conclude, that even at this temperature the character of the elemental distribution points to the mutual diffusion of components, though the interaction is still localized within a 50–70 nm zone. According to TEM studies combined with point-to point EDX analysis, in samples, processed at 1100 °C, large monolithic SrZrO₃ particles (continuous area with a clearly defined boundary on the right side of Fig. 4a) are embedded in the mass of disordered LSM crystallites (dark particles on the left side). The mean size of the latter ones being close to 50 nm, so that the interaction is probably limited by a one-crystallite layer of LSM, which covers the SrZrO₃ microcrystals, and by surface layers of SrZrO₃ particles. Similarly to an LSM–YSZ interaction,^{7,11,13} Zr diffusion into LSM can be neglected, and the basic feature of the process is the La and Mn out-diffusion into the SrZrO₃.

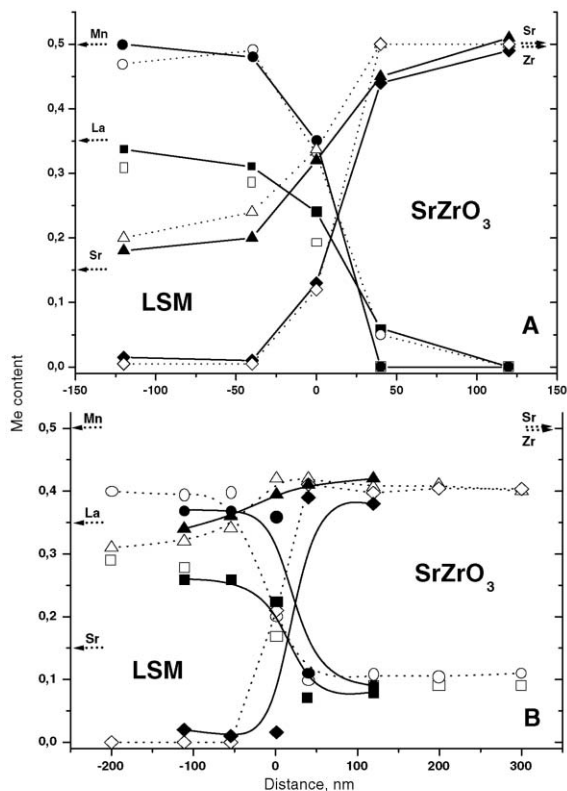


Fig. 3 Composition profiles of elements, measured during point-by-point EDX nanoprobe analysis of the interface between the LSM and SrZrO₃ particles in 80% LSM–20% SrZrO₃ composites. Annealing temperature: (A) 1100 °C; (B) 1300 °C. ○ and ●, Mn; ■ and □, La; ▲ and △, Sr; ◆ and ◇, Zr. Arrows show the values corresponding to a stoichiometric ratio of cations.

The character of the interaction can be clearly seen at 1300 °C, when the sample is almost paramagnetic and the SrZrO₃ lattice parameters are substantially changed compared with the initial phase. While at 1100 °C the elemental ratio outside the contact area is close to the stoichiometric values for both LSM and SrZrO₃ (Fig. 3a), at 1300 °C the chemical reaction results in a clearly detectable cation redistribution at least within a zone 200–300 nm from the reaction interface (Fig. 3b).

The most interesting and unusual result of this interaction is the appearance of La and Mn within the SrZrO₃ grains. The intensity of their EDX signals substantially exceeds the background level. To ensure statistical plausibility, the measurements were performed at several interaction boundaries at different points in the composite samples (the results of the two measurements are given as filled and empty symbols in Fig. 3a and 3b). The absence of visible gradients for the La and Mn concentrations within at least 300 nm of the reaction boundary makes a uniform distribution rather probable throughout the SrZrO₃ grains.

TEM analysis shows that annealing at 1300 °C is accompanied by crystallographic ordering and intensive grain growth of LSM particles (Fig. 4b). The component distribution within these grains is still not uniform; and chemical inhomogeneities are clearly visible in the vicinity of the reaction interface. Short sintering times are characteristic of the LSM materials,¹⁶ while further annealing usually leads to the formation of large grains with uniform composition.¹⁷ The interface for a topochemical reaction is blurred, so that a distinct boundary between LSM and SrZrO₃ crystallites is almost absent. At the same time the main reason for the elemental redistribution, easily observable even at 200 nm from the reaction boundary, is the La and Mn out-diffusion, which lead to a decrease in their concentrations, compared to stoichiometric values, and to a corresponding substantial increase in the Sr content over stoichiometry

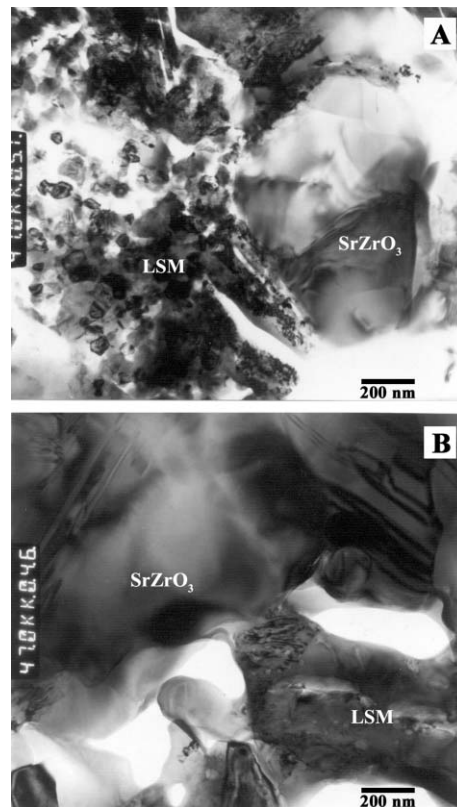


Fig. 4 TEM micrographs of the LSM composites (80% LSM–20% SrZrO₃) annealed at: (A) 1100 °C; (B) 1300 °C.

(triangles in Fig. 3b). The single-sided character of this diffusion can also be illustrated by the very low level of Zr EDX signal in the LSM particles, especially outside the area of intensive interaction.

Concluding remarks

The present investigation was aimed at seeking the chemical background for several unusual physical phenomena observed in CMR composites. At the same time the results obtained, which clearly reveal the substantial chemical reactivity of SrZrO₃ (even with a rather short reaction time and at moderately high temperatures (1200–1300 °C)), might be of interest for further studies into the processes occurring in SOFCs. The character of the processes occurring during the LSM–SrZrO₃ interaction is rather similar to the first stages of the solid-state reaction between LSM and YSZ. However, most of the microprobe studies of this reaction did not detect La or Mn within the SrZrO₃ particles. This discrepancy could be due to the higher reactivity of the finely grained LSM powders, used in this work for the preparation of the composite. Another reason might be related to the small size of the SrZrO₃ particles, obtained as an LSM–YSZ interaction product and which are comparable in size with the diameter of the microprobe excitation zone, a fact that made it difficult to correctly interpret the EDX results in this case. This last feature makes reasonable the wider application of nanoprobe in solid-state reaction studies, because, as has been demonstrated, some important processes cannot be detected when the usual micron-sized probe is used.

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